

# MoS<sub>2</sub> Polymorphic Engineering Enhances Selectivity in the Electrochemical Reduction of Nitrogen to Ammonia

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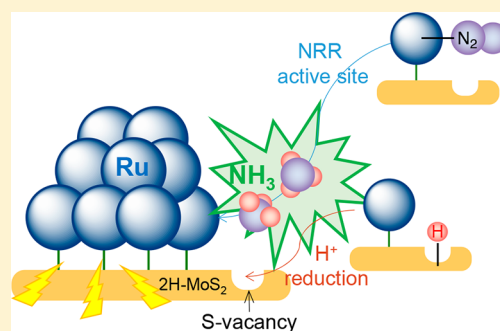
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## Supporting Information

**ABSTRACT:** The electrochemical N<sub>2</sub> reduction reaction (NRR) offers a direct pathway to produce NH<sub>3</sub> from renewable energy. However, aqueous NRR suffers from both low Faradaic efficiency (FE) and low yield rate. The main reason is the more favored H<sup>+</sup> reduction to H<sub>2</sub> in aqueous electrolytes. Here we demonstrate a highly selective Ru/MoS<sub>2</sub> NRR catalyst on which the MoS<sub>2</sub> polymorphs can be controlled to suppress H<sup>+</sup> reduction. A NRR FE as high as 17.6% and NH<sub>3</sub> yield rate of  $1.14 \times 10^{-10}$  mol cm<sup>-2</sup> s<sup>-1</sup> are demonstrated at 50 °C. Theoretical evidence supports a hypothesis that the high NRR activity originates from the synergistic interplay between the Ru clusters as N<sub>2</sub> binding sites and nearby isolated S-vacancies on the 2H-MoS<sub>2</sub> as centers for hydrogenation; this supports formation of NH<sub>3</sub> at the Ru/2H-MoS<sub>2</sub> interface.



It is reported that the conventional Haber–Bosch process consumes approximately 2% of global energy supply and contributes ~1.5% of global greenhouse gas emissions.<sup>1</sup> Renewable energy-powered dinitrogen (N<sub>2</sub>) fixation technologies to synthesize ammonia (NH<sub>3</sub>) offer a core solution to the energy and environmental challenges caused by this conventional Haber–Bosch process.<sup>2</sup>

The electrochemical nitrogen reduction reaction (NRR) at ambient conditions offers a direct pathway for the conversion of renewable electricity into NH<sub>3</sub> in a simple electrolytic cell.<sup>3</sup> The required H<sup>+</sup> can be sustainably generated from a water oxidation process.<sup>4</sup> However, the NRR process is kinetically sluggish, involving the addition of 6e<sup>-</sup> and 6H<sup>+</sup>, and is electrochemically disfavored over the more facile 2e<sup>-</sup> and 2H<sup>+</sup> hydrogen evolution reaction (HER) in aqueous solution.<sup>3,5</sup> The highly disruptive HER largely explains why most of the recently reported metal-based electrocatalysts such as Ru,<sup>6</sup> Pt,<sup>7</sup> Au,<sup>8</sup> Rh,<sup>9</sup> Fe,<sup>10,11</sup> Mo,<sup>12</sup> Pd,<sup>13</sup> Cr,<sup>14</sup> and N-doped carbon-based materials<sup>15,16</sup> often suffer from both low NRR Faradaic efficiency (FE) of <15% and low NH<sub>3</sub> yield rates in the range of 10<sup>-11</sup>–10<sup>-10</sup> mol cm<sup>-2</sup> s<sup>-1</sup>.

Recent efforts are devoted toward the development of NRR electrocatalysts based on rational approaches.<sup>3</sup> Common

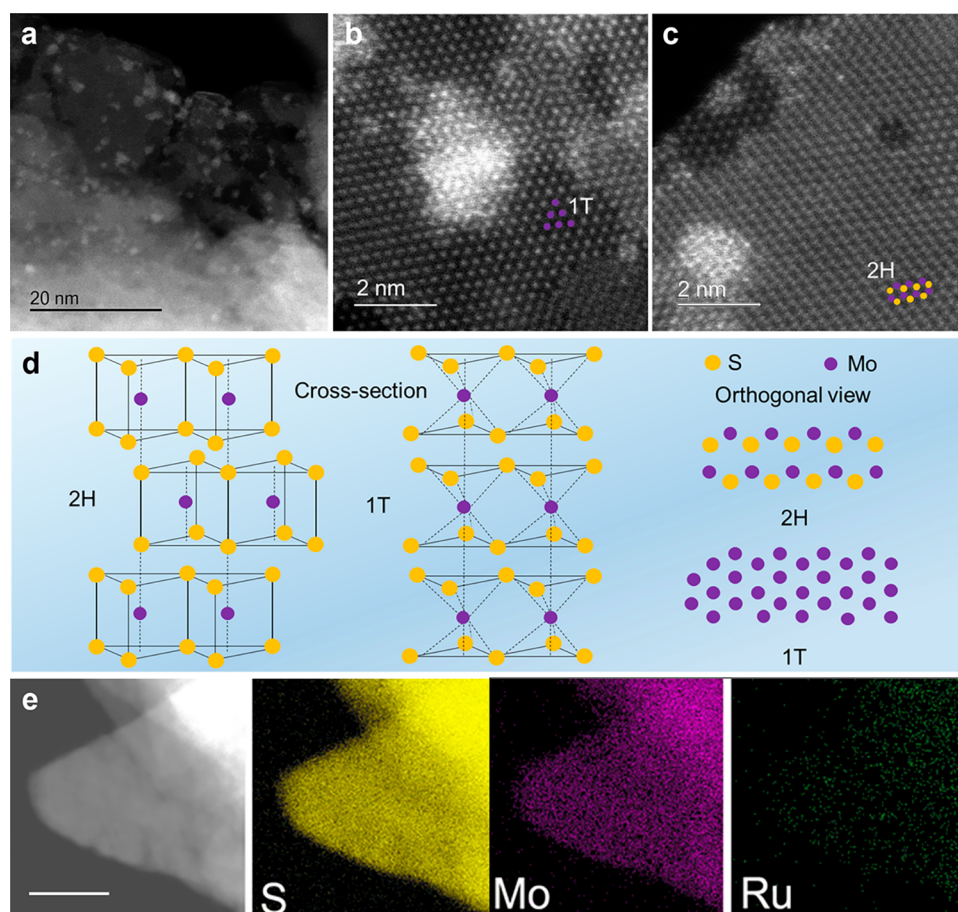
strategies are the design of inherently active NRR catalysts guided by theoretical calculations and inspired by nature's nitrogenase enzymes.<sup>17–20</sup> Another design approach is the suppression of HER rates. Because the initial H<sup>+</sup> adsorption to catalytically active sites (\*) in the HER process is driven by charge transfer (H<sup>+</sup> + e<sup>-</sup> + \* ↔ \*H), it is proposed that the physical properties of the catalyst, substrate, and electrolyte are crucial in limiting HER.<sup>3,21</sup> Our group has proposed novel strategies including the use of aprotic<sup>22</sup> and ionic liquids<sup>23</sup> as electrolytes to limit HER.

In the present Letter, we demonstrate the impact of HER suppression on the NRR activities of a composite catalyst structure. The MoS<sub>2</sub>, which is known for having tunable HER activities, is decorated with Ru clusters to provide binding sites for N<sub>2</sub> activation. NRR activity and selectivity is significantly enhanced on “semiconducting” type Ru-decorated 2H-MoS<sub>2</sub> (denoted as Ru/2H-MoS<sub>2</sub>), in contrast to that of “metallic” type 1T-MoS<sub>2</sub>. It is demonstrated that the ability to control

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**Figure 1.** (a) High-angle annular dark field (HAADF) scanning transmission electron microscopy (STEM) images of Ru/2H-MoS<sub>2</sub>; high-resolution HAADF-STEM imaging of (b) Ru/1T-MoS<sub>2</sub> and (c) Ru/2H-MoS<sub>2</sub>; (d) schematic illustration of the crystallographic difference between 2H and 1T-MoS<sub>2</sub>; and (e) energy-dispersive X-ray (EDS) elemental mapping (scale bar = 125 nm).

HER kinetics by tuning MoS<sub>2</sub> properties via polymorphic engineering is critical in achieving high NRR activity.

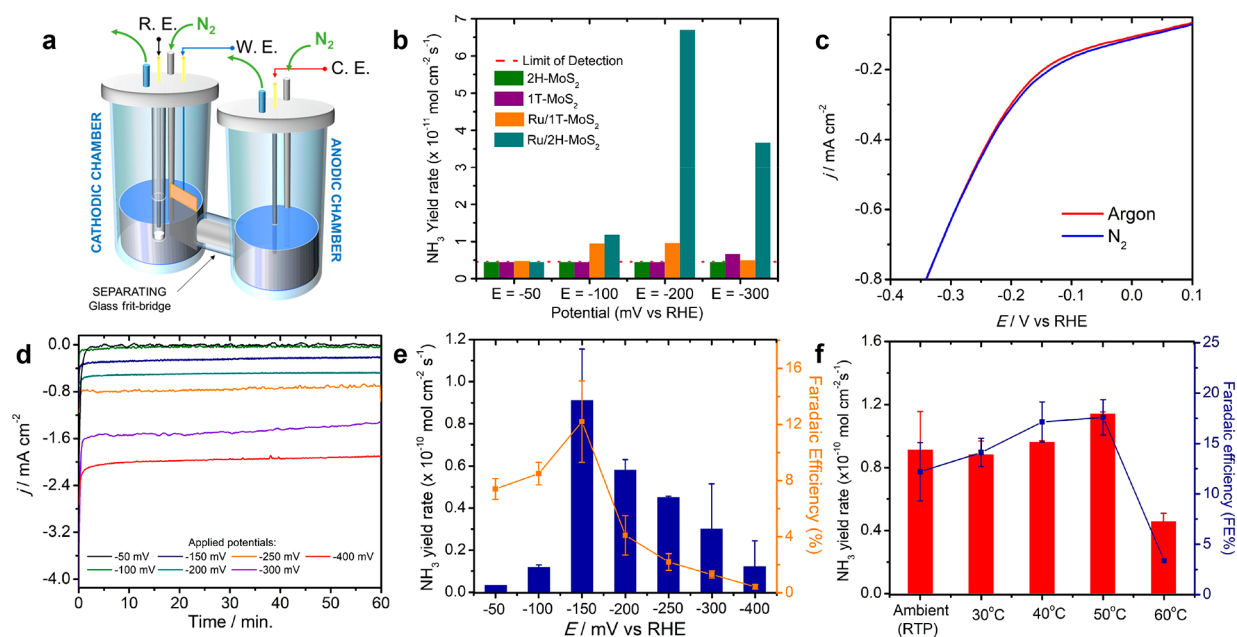
Scanning transmission electron microscopy (STEM) images (Figures 1a and S1) at low magnifications show the typical morphology of the synthesized Ru/MoS<sub>2</sub>. The high-resolution images in Figure 1b,c show that the bright Ru clusters are amorphous (see section 2.1 in the Supporting Information for further discussion). The 1T-MoS<sub>2</sub> phase transformation to 2H-MoS<sub>2</sub> following hydrothermal treatment was evident in the images, as visualized by their crystallographic patterns, Figure 1d, superimposed on Figure 1b,c (see section 2.1 in the Supporting Information for further discussion). Furthermore, energy-dispersive X-ray (EDS) characterization shown in Figures S2 and 1e validates the presence of Ru. Additionally, physical characterizations including X-ray diffraction (XRD) and Raman spectroscopy have been carried out and are discussed in section 2.1 of the Supporting Information (Figure S3).

High-resolution X-ray photoelectron spectroscopy (XPS) of Mo 3d and Ru 3p (Figure S4) clearly demonstrates the polymorphic transformation process of MoS<sub>2</sub> induced by hydrothermal treatment, while the Ru clusters remain unchanged. Important to note are that the S:Mo atomic ratios observed in both Ru/1T-MoS<sub>2</sub> and Ru/2H-MoS<sub>2</sub> were found to be  $\sim 1.65$ , indicating the presence of S-vacancies in the Ru/MoS<sub>2</sub>, which is also confirmed with HAADF-STEM shown in Figure S5.<sup>24</sup> The important role of S-vacancies in the initial

step of proton absorption of HER has been widely demonstrated in prior literature. Such H<sup>+</sup>-S-vacancy structures can dissociate to hydrogen atoms, H<sup>\*</sup>, to provide the six proton-coupled electron transfer (PCET) steps of the NRR,<sup>25–27</sup> which will be explained by the density functional theory calculation (DFT, vide infra).

Initially, the NRR activity of each material was evaluated using a H-cell with a standard three-electrode setup (Figure 2a; further details are provided in section 1.1 of the Supporting Information) using controlled potential electrolysis (CPE) experiments. As shown in Figure 2b, Ru/2H-MoS<sub>2</sub> exhibits the highest NH<sub>3</sub> yield rate of  $6.7 \times 10^{-11}$  mol cm<sup>-2</sup> s<sup>-1</sup>, almost an order of magnitude higher than that of Ru/1T-MoS<sub>2</sub> ( $9.6 \times 10^{-12}$  mol cm<sup>-2</sup> s<sup>-1</sup> at  $-200$  mV). The NRR performances of undecorated 2H-MoS<sub>2</sub> and 1T-MoS<sub>2</sub> across the tested potentials are generally very low, with most indophenol tests indicating results comparable to that of the background (Figure S8).

Voltammetric verification of NRR on Ru/2H-MoS<sub>2</sub> was carried out by linear sweep voltammetry (LSV). As shown in Figure 2c, a slightly higher cathodic current density is observed within the sweeping potential range of  $-50$  to  $-200$  mV vs RHE under N<sub>2</sub> gas purging compared to that of Ar gas purging. Furthermore, a N<sub>2</sub>-Ar gas switching experiment (Figure S9) conducted at 150 mV vs RHE shows higher cathodic currents by about  $\sim 14\%$  when the gas stream was switched from Ar to N<sub>2</sub>. An extensive set of control experiments including



**Figure 2.** (a) Schematic diagram of the electrochemical cell used for NRR. (b) NRR activity of 2H-MoS<sub>2</sub>, 1T-MoS<sub>2</sub>, Ru/1T-MoS<sub>2</sub>, and Ru/2H-MoS<sub>2</sub> in 10 mM aqueous HCl. (c) Linear sweep voltammograms (LSVs) of Ru/2H-MoS<sub>2</sub> in the presence of nitrogen and argon in 10 mM HCl collected with planar glassy carbon electrode. (d) Corresponding chronoamperometric ( $j-t$ ) curves, (e) NRR performance (NH<sub>3</sub> yield rate and Faradaic efficiency) vs potential (mV), and (f) NRR performance at different electrolysis temperatures at a constant applied potential of  $-150$  mV vs RHE.

quantitative <sup>15</sup>N<sub>2</sub> (Figure S10) were carried out as described in section 2.2 of the Supporting Information.

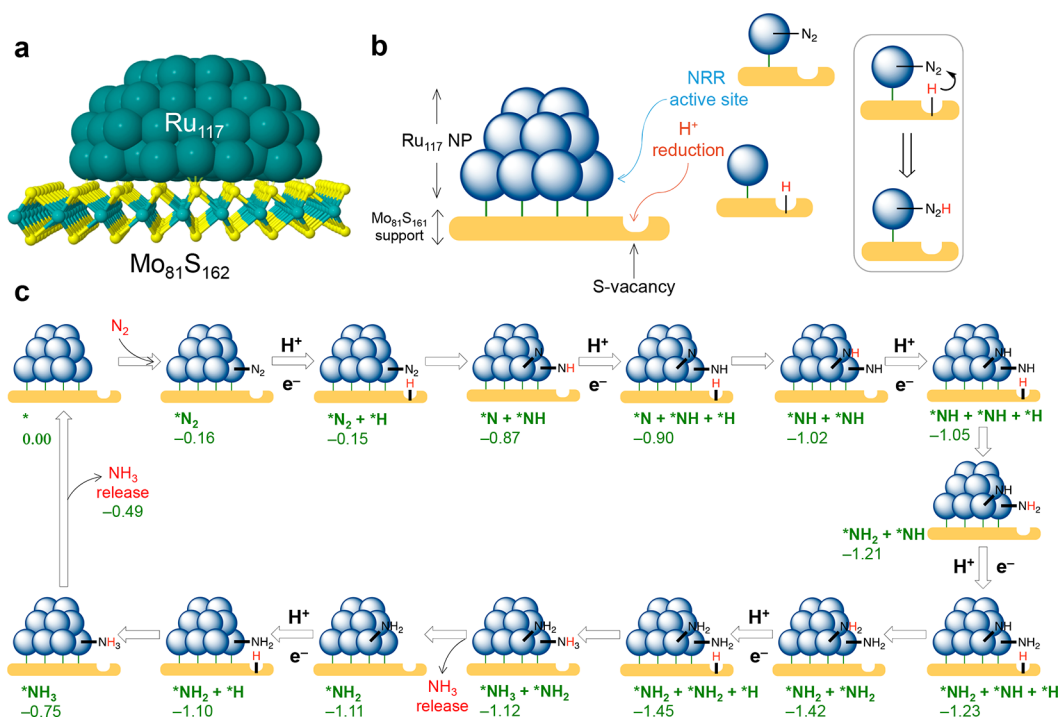
The  $j-t$  curve in Figure 2d shows the current responses obtained at several cathodic potentials between  $-50$  and  $-400$  mV vs RHE. Figure 2e shows that a NRR selectivity (FE) of  $8.5 \pm 0.8\%$  at a NH<sub>3</sub> yield rate of  $1.2 \pm 0.1 \times 10^{-11}$  mol cm<sup>-2</sup> s<sup>-1</sup> can be obtained at  $-100$  mV vs RHE. Consistent with a rapidly increasing rate of NRR compared to HER in this region of the potential, the application of a more negative potential of  $-150$  mV increases the FE considerably to  $12.2 \pm 3.0\%$  (data is the mean and standard deviation from  $n = 4$  experiments). In addition, no detectable amount of hydrazine was detected after typical CPE at  $-150$  mV (Figure S11). At this potential, an NH<sub>3</sub> yield rate of  $9.1 \pm 0.2 \times 10^{-11}$  mol cm<sup>-2</sup> s<sup>-1</sup> was measured. Furthermore, the NRR at  $-150$  mV was confirmed by replacing the purging gas with Ar. Following 1 h of CPE, the detected ammonia was comparable to that of the background values ( $2.4 \pm 0.2$  nmol ml<sup>-1</sup>,  $n = 2$  experiments, Figure S12). Note that this amount has been subtracted from all quantitative results (NH<sub>3</sub> yield rate and NRR FE) reported later in this study as a background value. At potentials more negative than  $-300$  mV vs RHE, the increased dominance of HER results in significantly increased current density, resulting in concomitant loss of NRR FE and NH<sub>3</sub> yield rate. This phenomenon was further verified using gas chromatography measurement of the produced H<sub>2</sub> (Figure S13).

The effect of temperature on NRR performance was also investigated (Figures 2f and S15) at the optimized potential. As shown in Figure 2f, slight increases in the FE and NH<sub>3</sub> yield rate to 17.6% and  $1.14 \times 10^{-10}$  mol cm<sup>-2</sup> s<sup>-1</sup> were observed at an electrolysis temperature of 50 °C. Increasing the NRR electrolysis temperature further to 60 °C resulted in a significant decrease in both the FE and NH<sub>3</sub> yield rate, as expected due to the decreased N<sub>2</sub> solubility and increased HER kinetics in the electrolyte at higher temperatures. Comparing

the Ru/2H-MoS<sub>2</sub> NRR performance with literature data (Table S3 and Figure S15), it appears that Ru/2H-MoS<sub>2</sub> is among the most efficient in the field (Figure S15).

Additionally, the Ru/2H-MoS<sub>2</sub> is able to maintain relatively stable NRR catalytic activity, as indicated by the stable current profile in Figure S16. Once steady state is established after  $\sim 30$  min, the yield rises linearly in time (Figure S17b) during 4 h continuous electrolysis at  $-150$  mV vs RHE; after the fourth hour, a total of  $622 \pm 34$  nmols of NH<sub>3</sub> is recovered. The FE in the steady-state region of Figure S17 was calculated to be 13.8% with a yield rate of  $0.46 \times 10^{-10}$  mol cm<sup>-2</sup> s<sup>-1</sup>. A set of physical characterizations, including Raman spectroscopy (Figure S18), SEM-EDS (Figure S19), and inductively coupled plasma–optical emission spectrometry (ICP-OES, Table S2), were also conducted to verify the physical stability of the catalyst.

LSV was also used to establish the polymorphic correlation of MoS<sub>2</sub> to the HER activity for each of the materials. Figure S20 shows that Ru/1T-MoS<sub>2</sub> exhibits a HER onset overpotential of 49 mV and a Tafel slope of  $\sim 66$  mV dec<sup>-1</sup>. In contrast, the LSVs of 1T-MoS<sub>2</sub> reveal an onset overpotential of 233 mV and a Tafel slope of  $\sim 153$  mV dec<sup>-1</sup>. Similar trends in HER activities were also observed in the 2H-MoS<sub>2</sub>-based samples (phase-transformed); 2H-MoS<sub>2</sub> exhibits an onset overpotential of 330 mV, while Ru/2H-MoS<sub>2</sub> exhibits an onset overpotential of 187 mV and a Tafel slope of  $\sim 145$  mV dec<sup>-1</sup>. Therefore, the form of the MoS<sub>2</sub> critically influences the HER kinetics, both with and without Ru nanoparticle decoration. One major factor that contributes greatly to this phenomenon is the semiconductivity of 2H-MoS<sub>2</sub>, compared to that of 1T-MoS<sub>2</sub>.<sup>28</sup> Because the H<sup>+</sup>/H<sub>2</sub> redox potential lies above the conduction band (CB) energy level of 2H-MoS<sub>2</sub>, it has the effect of inhibiting the rate of HER, as is well-known.<sup>29</sup> Hence, the phase-dependent HER activity behavior of MoS<sub>2</sub> can be utilized to promote NRR to become more competitive



**Figure 3.** In silico design and electronic properties of Ru/2H-MoS<sub>2</sub> and DFT analysis for the NRR mechanism. (a) Initial model consisting of a half-spherical hcp Ru<sub>117</sub> nanocluster on a perfect 9 × 9 2H-MoS<sub>2</sub> (Mo<sub>81</sub>S<sub>162</sub>) pristine supercell. (b) We hypothesize that NRR is catalyzed by Ru/MoS<sub>2</sub> in a synergetic process in which the Ru moiety acts as a proper NRR active site while the S-vacancy on pristine behaves as a HER active site. Once N<sub>2</sub> is adsorbed, a reduced proton on a S-vacancy will subsequently provide hydrogen to hydrogenate the N<sub>2</sub>. (c) Schematic minimum-energy pathway for electrochemical N<sub>2</sub> conversion into NH<sub>3</sub> catalyzed by Ru/2H-MoS<sub>2</sub> material. Relative Gibbs free energies are shown in eV at mild conditions (RPBE functional) for the dissociative pathway when there is no applied bias ( $U = 0$  V) and pH = 0. Note: “\*” denotes adsorbed species.

against the HER. It is also worth noting that this observation is consistent with the behavior of some previously reported NRR catalysts containing semiconducting materials such as Fe<sub>2</sub>O<sub>3</sub>, BiVO<sub>4</sub>, CeO<sub>x</sub>, and TiO<sub>2</sub>.<sup>10,30–32</sup>

DFT calculations were carried out with an optimized heterostructure and interpretative model (Figure 3a,b; further discussions are provided in section 2.4 of the Supporting Information), where hydrogenated S-vacancies provide a fundamental role in the NRR mechanism as the H-provider because the already formed \*H can be transferred directly to nearby bound N<sub>2</sub> or to the nitrogen reduction intermediates, for example  $*_{\text{Ru}}\text{N}_2 + *_{\text{S-vac}}\text{H} \rightleftharpoons *_{\text{Ru}}\text{N}_2\text{H}$ , with the regeneration of the corresponding S-vacancy for further proton reduction steps. Contrary to what has been described for similar materials,<sup>22,33</sup> adsorbed N<sub>2</sub>H is not a stable species in the minimum-energy path when catalyzed by Ru (Figure 3 midtop). Instead, the hydrogenation of \*N<sub>2</sub> produces, in this case, spontaneous cleavage into \*N and \*NH, being 0.72 eV lower in energy with respect to \*N<sub>2</sub> + \*H.<sup>34</sup>

Once \*N + \*NH are formed (Figure 3 top), a set of successive hydrogenations occur, in which hydrogen is provided from the S-vacancy. Specifically, the analysis of the minimum-energy pathway reveals the preferred formation of the \*NH + \*NH, \*NH<sub>2</sub> + \*NH, \*NH<sub>2</sub> + \*NH<sub>2</sub>, and \*NH<sub>3</sub> + \*NH<sub>2</sub> species during the second to fifth hydrogenations of the nitrogen intermediates. Generation of the first adsorbed ammonia molecule at \*NH<sub>3</sub> + \*NH<sub>2</sub> seems to be the rate-limiting step of the whole process, showing a thermodynamic impediment of +0.33 eV. The final hydrogenation of the remaining \*NH<sub>2</sub> also exhibits a similar thermodynamic impediment of +0.35 eV.

In conclusion, the present work highlights the importance of catalyst design in achieving a highly active and selective NRR electrocatalyst. It is demonstrated that the semiconducting property of 2H-MoS<sub>2</sub> plays an important role in suppressing the HER rate and therefore improving the NRR selectivity of the Ru/2H-MoS<sub>2</sub>. The electrocatalyst exhibits a high FE of 17.6% and NH<sub>3</sub> yield rate of  $1.14 \times 10^{-10}$  mol cm<sup>-2</sup> s<sup>-1</sup> at 50 °C, rendering Ru/2H-MoS<sub>2</sub> among the most efficient in the field (Figure S15). Both experimental and theoretical evidence suggest that the NRR is catalyzed at the interface of the Ru clusters as the active N<sub>2</sub> binding sites and the S-vacancy in the 2H-MoS<sub>2</sub> as the active H<sup>+</sup> binding sites. The strategy of using semiconducting materials could therefore be further extended in the research and development of catalytic materials for other reactions that are limited by the HER, such as carbon dioxide reduction.

## ■ ASSOCIATED CONTENT

### Supporting Information

The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acsenergylett.8b02257.

Experimental, TEM, XRD, XPS, and Raman characterizations, control experiments, <sup>15</sup>N<sub>2</sub> experiments, hydrazine test, gas chromatography, HER performance of Ru/MoS<sub>2</sub>, supplementary figures, table of performance, and computational details (PDF)

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## Notes

The authors declare no competing financial interest.

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